

# Modeling of Microstructure Evolution During LENS™ Deposition

Liang Wang, PhD

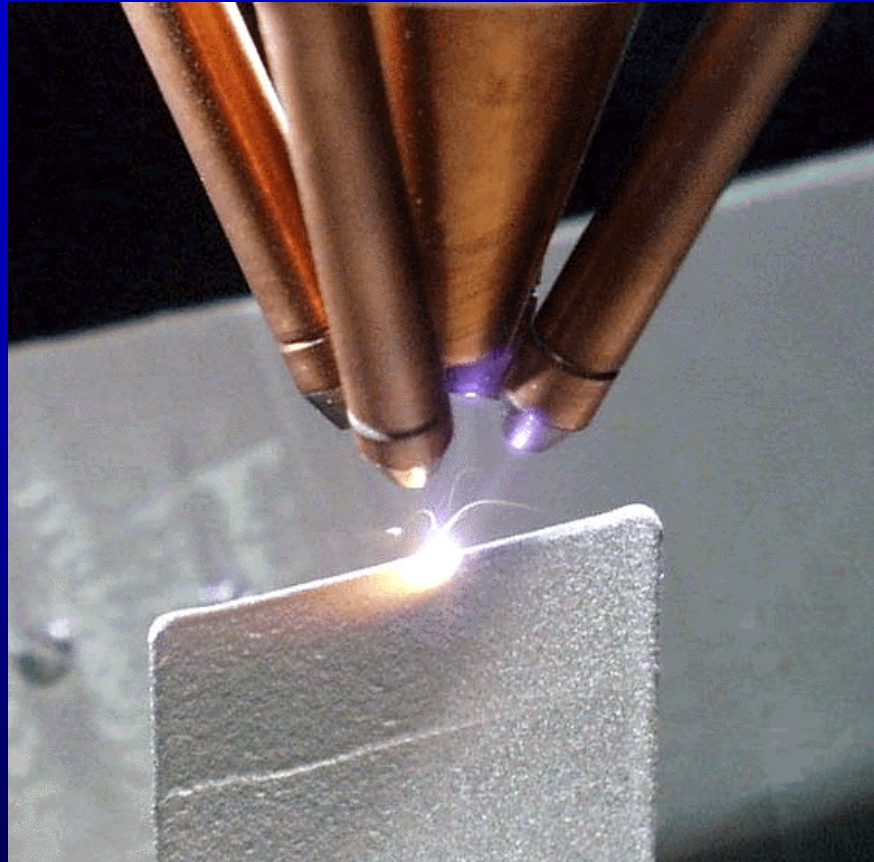
Haitham El Kadiri, PhD

Sergio Felicelli, PhD

Mark Horstemeyer, PhD

Paul Wang, PhD

Center for Advanced Vehicular Systems  
Mississippi State University



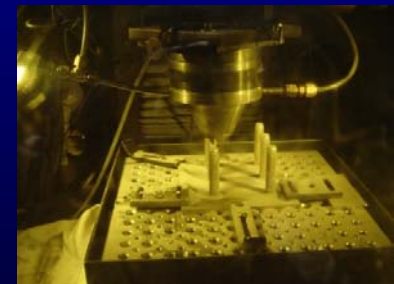
## Laser Engineered Net Shaping (LENS)

# Introduction

- A variety of materials can be used:
  - Steel Materials (4140, SS410, SS316)
  - Ti-based alloy (Ti-6Al-4V)
  - Inconel, copper, aluminum, etc.
- Application:
  - Aerospace repair & overhaul
  - Rapid prototyping and 3D structure fabrication
  - Product development for aerospace, defense, and medical markets, etc.
- Advantages:
  - Low cost & time saving
  - Enhanced design flexibility and automation
  - Highly localized heat-affected zone (HAZ)
  - Superior material properties (strength and ductility)



Processing Blade



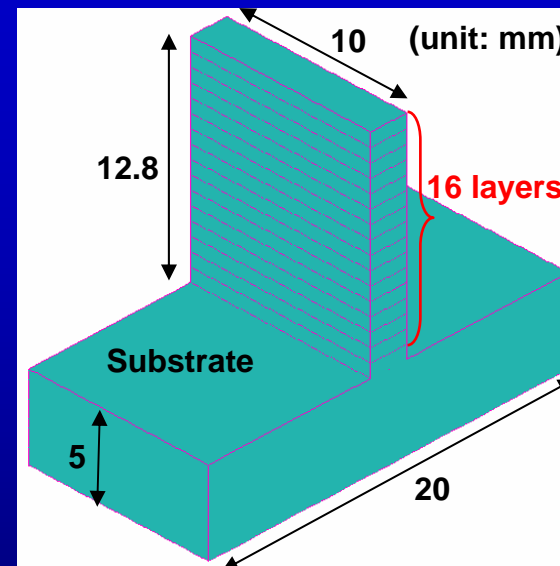
Processing Bar

# Objectives

- Develop a 3-D thermal-metallurgical model to simulate 16-pass single build plate LENS deposition of 4140 steel powder with SYSWELD finite element code.
- Predict the thermal profiles, phase transformation, and hardness in the deposited part, and compared with experimental data
- Investigate the effect of the thermal cycles on the phase transformation and consequent hardness.

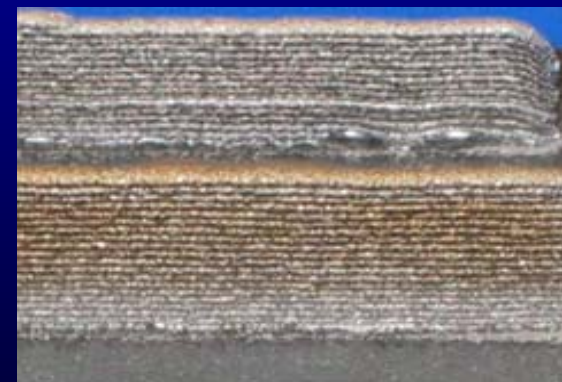
# Geometry & Process Parameters

Process parameters	Values
Width of the part	2.0mm
Thickness for each layer	0.8mm
Laser beam travel velocity	8.5mm/s
Moving time of the laser beam for each pass	1.18s
Idle time of consecutive layers deposition	0.32s
Time to finish one layer	1.5s
Total time to finish the part	24s



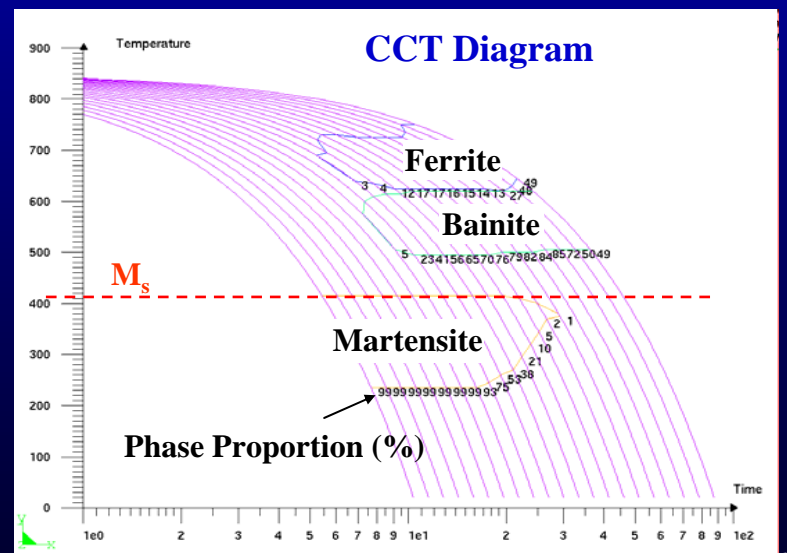
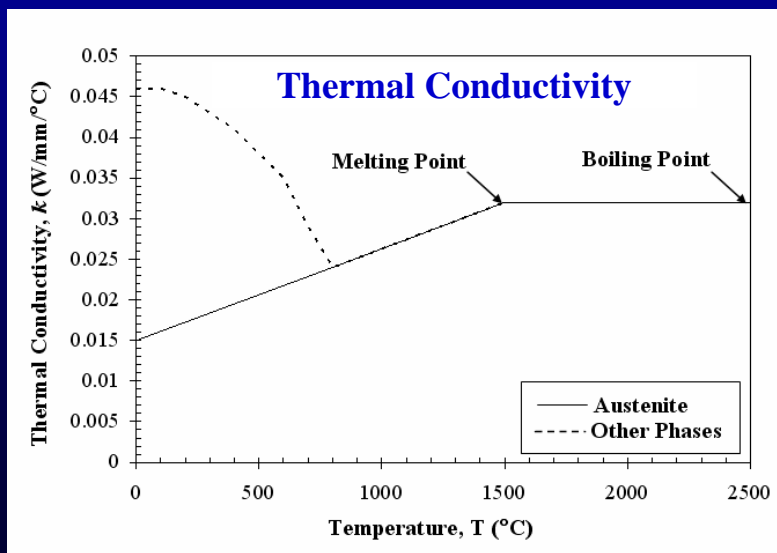
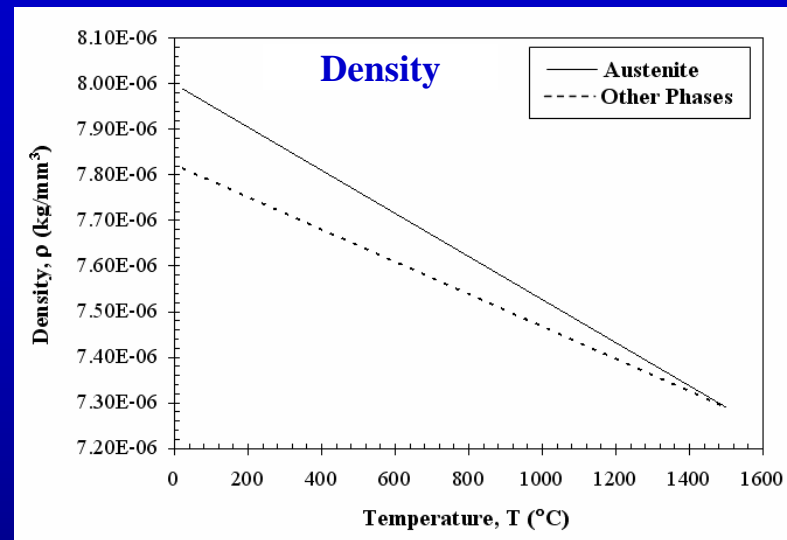
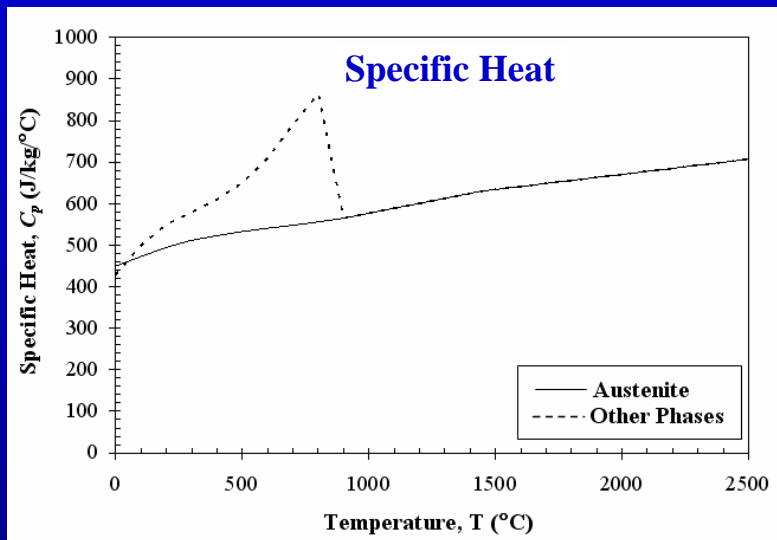
## Chemical composition of 4140 steel (wt%)

C	Si	Mn	Cr	Mo	P	S
0.38/0.43	0.15/0.30	0.75/1.0	0.8/1.1	0.15/0.35	<0.035	<0.04



- Weld direction: Same direction for each pass
- Both the deposited part and the substrate are 4140 steel

# Thermal Properties (4140)



# Thermal-Metallurgical Model

- Modified heat conduction equation:

$$\left( \sum_i f_i (\rho C)_i \right) \frac{\partial T}{\partial t} - \nabla \cdot \left( \left( \sum_i f_i \lambda_i \right) \nabla T \right) + \sum_{i < j} L_{ij}(T) \cdot A_{ij} = Q$$

$f$  - phase proportion

$T$  - temperature

$t$  - time

$i, j$  - phase indexes

$L_{ij}(T)$  - latent heat of  $i \rightarrow j$  transformation

$A_{ij}$  - proportion of phase  $i$  transformed to  $j$  in time unit

$\rho$  - mass density

$C$  - specific heat

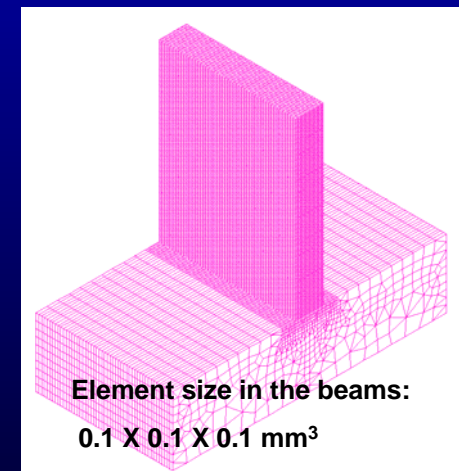
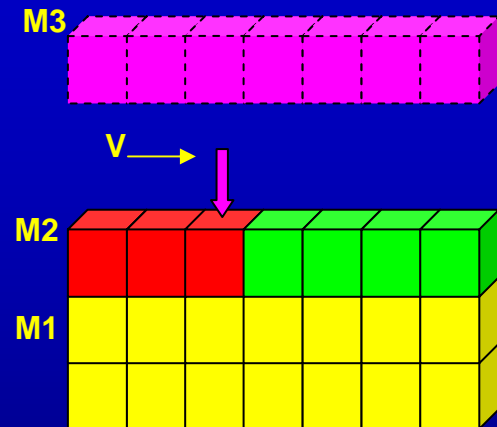
$\lambda$  - thermal conductivity

$Q$  - heat source



# Element Activation Technique

- Dummy material method is applied to the element activation:
  - M1: Deposited layers + substrate
    - Material with actual thermal properties and phase transformation
  - M2: Layer being deposited
    - Material with actual thermal properties and starting with dummy phase
    - Dummy phase  $\rightarrow$  Austenite phase ( $T > T_{\text{aus}}$ )
  - M3: Layers to be deposited
    - Material with dummy low thermal properties and without phase transformation
- Fixed mesh is used for the plate and substrate.





# Heat Source

- 3D Conical Gaussian Function

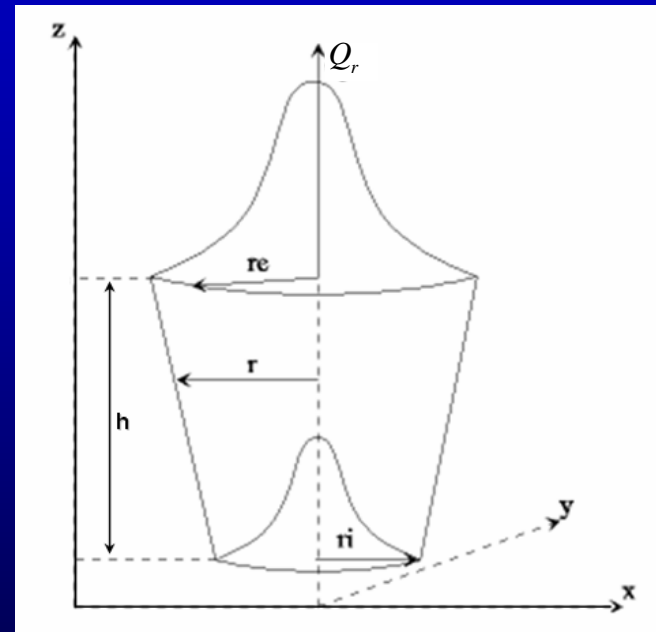
$$Q_r = \frac{2P}{\pi r_0^2 h} \left(1 - \frac{z}{h}\right) \exp\left(1 - \left(\frac{r}{r_0}\right)^2\right)$$

$$r_o = r_e - \frac{(r_e - r_i)(h - z)}{h}$$

$$r^2 = (x - x_o)^2 + (y - y_o - v \cdot t)^2$$

$Q_r$  - Input energy density (W/mm<sup>3</sup>)

$P$  - Absorbed laser power (W)



# Initial and Boundary Conditions

- Initial condition

$$T(x, y, z, t = 0) = T_0$$

- Boundary condition on the bottom of the substrate

$$T(x, y, z = 0) = T_0 \quad \text{for } t > 0$$

- Boundary conditions for all other surface

$$k(\nabla T \cdot \vec{n})|_{\Omega} = h(T - T_a)|_{\Omega} + \varepsilon\sigma(T^4 - T_e^4)|_{\Omega} - Q_r|_{\Omega Laser}$$

- As new layers are activated, the surfaces are increased and the boundary conditions are updated.

# Model for Phase Transformation

- Evolution equation for austenitic, ferritic-perlitic, and bainitic transformation of steels:

$$\frac{df_j}{dt} = \frac{\bar{f}_{j_{eq}}^{ij}(T) - f_j(T)}{\tau^{ij}(T)}$$

$f_j$  - proportion of phase  $j$  and  $\sum_j f_j = 1$

$\bar{f}_{j_{eq}}^{ij}(T)$  - equilibrium fraction of phase  $j$  at temperature  $T$

$\tau^{ij}(T)$  - characteristic time of the transformation (from  $i$  to  $j$ ) at temperature  $T$

These functions can be obtained by comparison of the prediction provided by the model with experimental results (i.e. CCT diagram).

# Martensitic Phase Transformation

- Martensitic transformation by Koistinen-Marburger Law:

$$f_m(T) = f_\gamma (1 - \exp(-0.011(M_s - T))) \quad \text{for } T \leq M_s$$

$f_\gamma$  - phase proportion of austenite before cooling down to  $M_s$

$M_s$  - initial transformation temperature

# Hardness Calculation

- **Hardness of ferrite (229 HV)** (Blondeau et al., 1973)

$$H_F \text{ (HV)} = 42 + 223C + 53Si + 30Mn + 12.6Ni + 7Cr + 1Mo \\ + \log V_r(10 - 19Si + 4Ni + 8Cr + 130V)$$

- **Hardness of bainite (337 HV)** (Blondeau et al., 1973)

$$H_B \text{ (HV)} = -323 + 185C + 330 Si + 153Mn + 65Ni + 144Cr + \\ 191Mo + \log V_r(89 + 53C - 55Si - 22Mn - 10Ni - 20Cr - 33Mo)$$

- **Hardness of martensite ( $H_{M0} = 800 \text{ HV}$ )** (Miokovic et al., 2006)

- **Hardness of tempered martensite** (Costa et al., 2005)

$$H_M = H_{M0} - A \left\{ \int_{t_1}^{t_2} \exp\left(-\frac{Q}{RT(t)}\right) dt \right\}^m$$

- $t_1$  is tempered martensite start time (the beginning of the thermal cycle in which martensite is tempered)
- $t_2$  is the final time when the part cools down to room temperature

# Hardness Calculation

- For each thermal cycle
  - If  $T_{max} > A_{c3}$ , then complete austenization
  - If  $T_{max} < A_{c1}$ , then martensite tempered
- If there are  $n$  thermal cycles, the proportion of retained austenite in the final part:

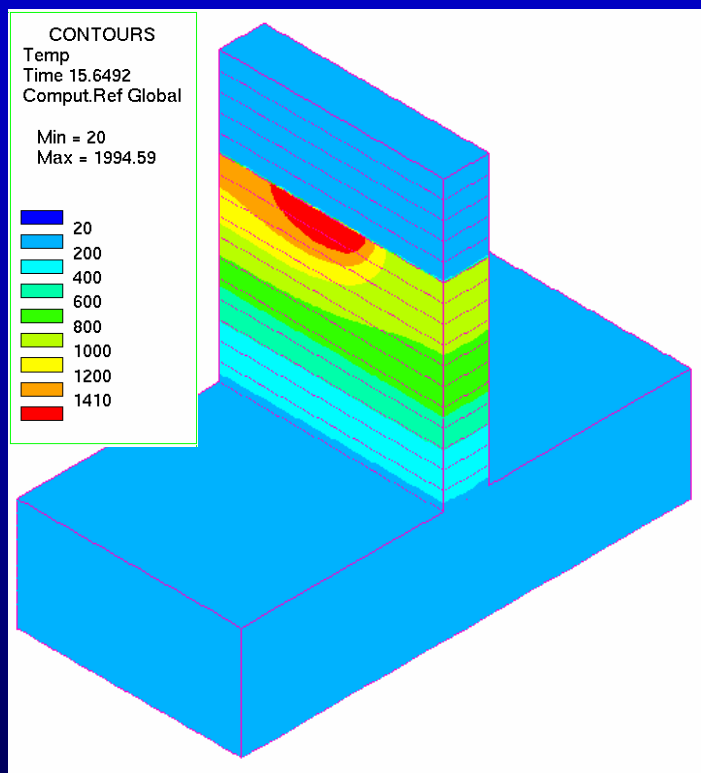
$$f_{\gamma} = f_{\gamma 0} \eta_1 \eta_2 \eta_3 \cdots \eta_n$$

$$\eta_i = \exp\left(-\frac{(M_s - T)}{90.9}\right) \quad (i = 1, 2, \dots, n)$$

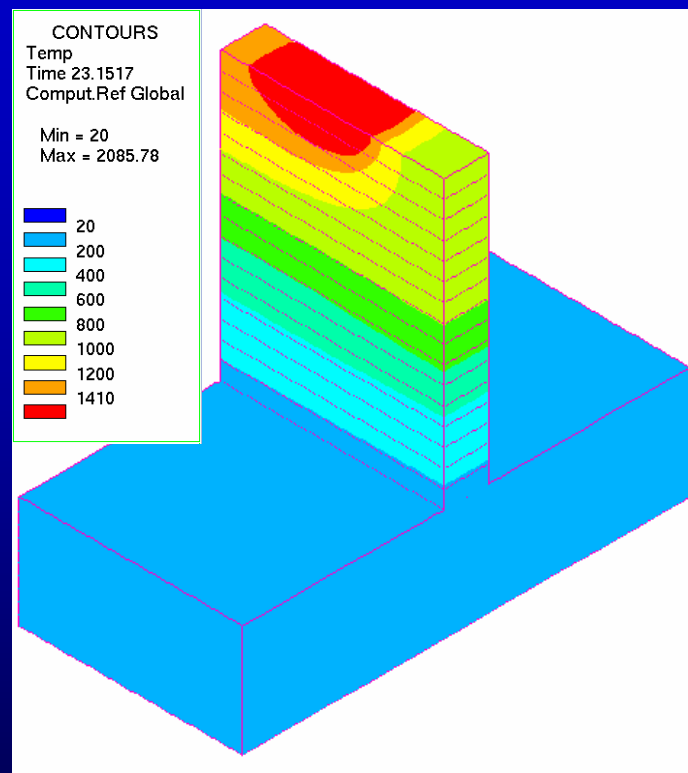
- **Hardness of the material:**

$$H = f_F \cdot H_F + f_B \cdot H_B + f_M \cdot H_M + f_{M0} \cdot H_{M0} + f_{\gamma} \cdot H_{\gamma}$$

# Temperature Contours



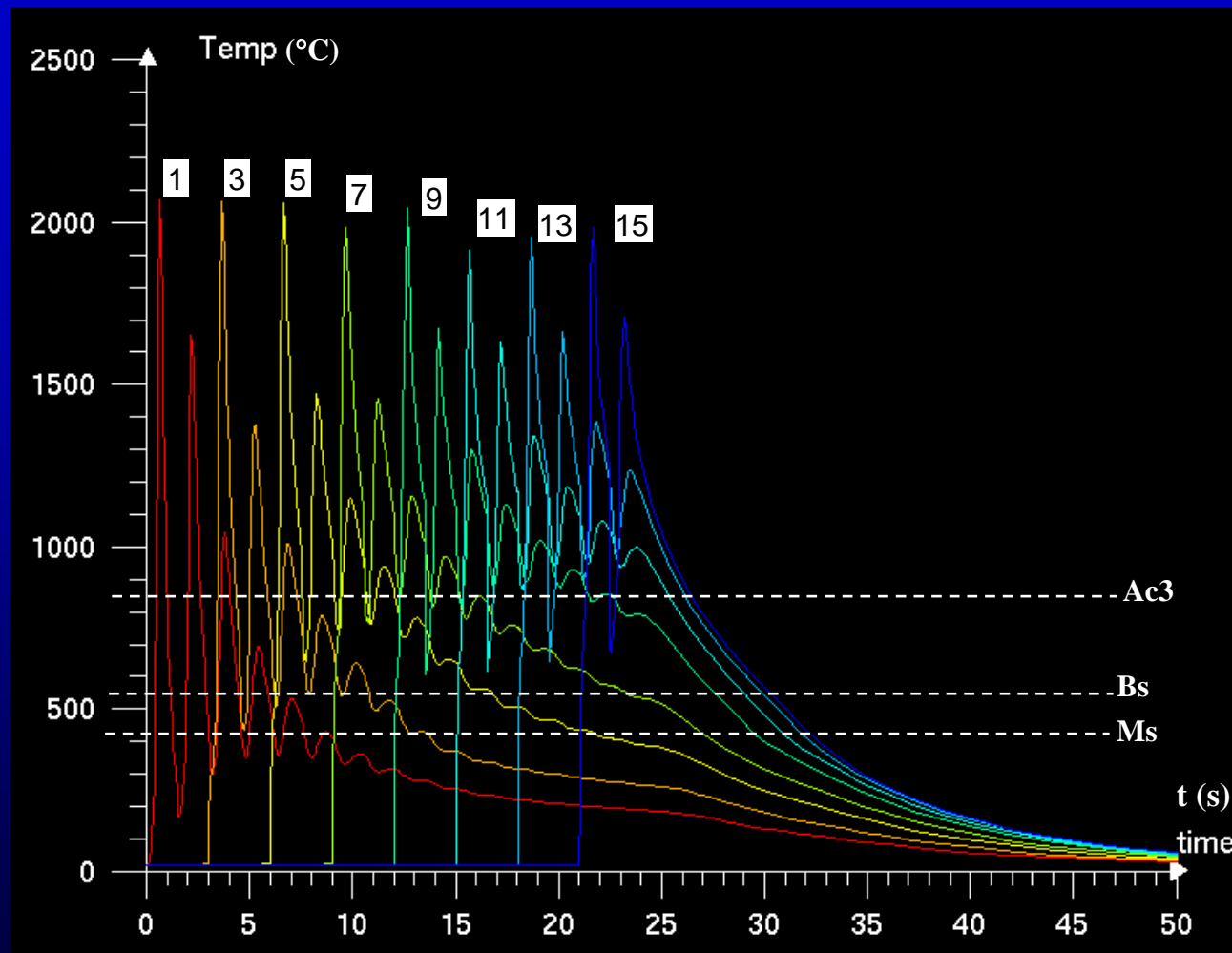
Deposition of 11<sup>th</sup> layer



Deposition of 16<sup>th</sup> layer

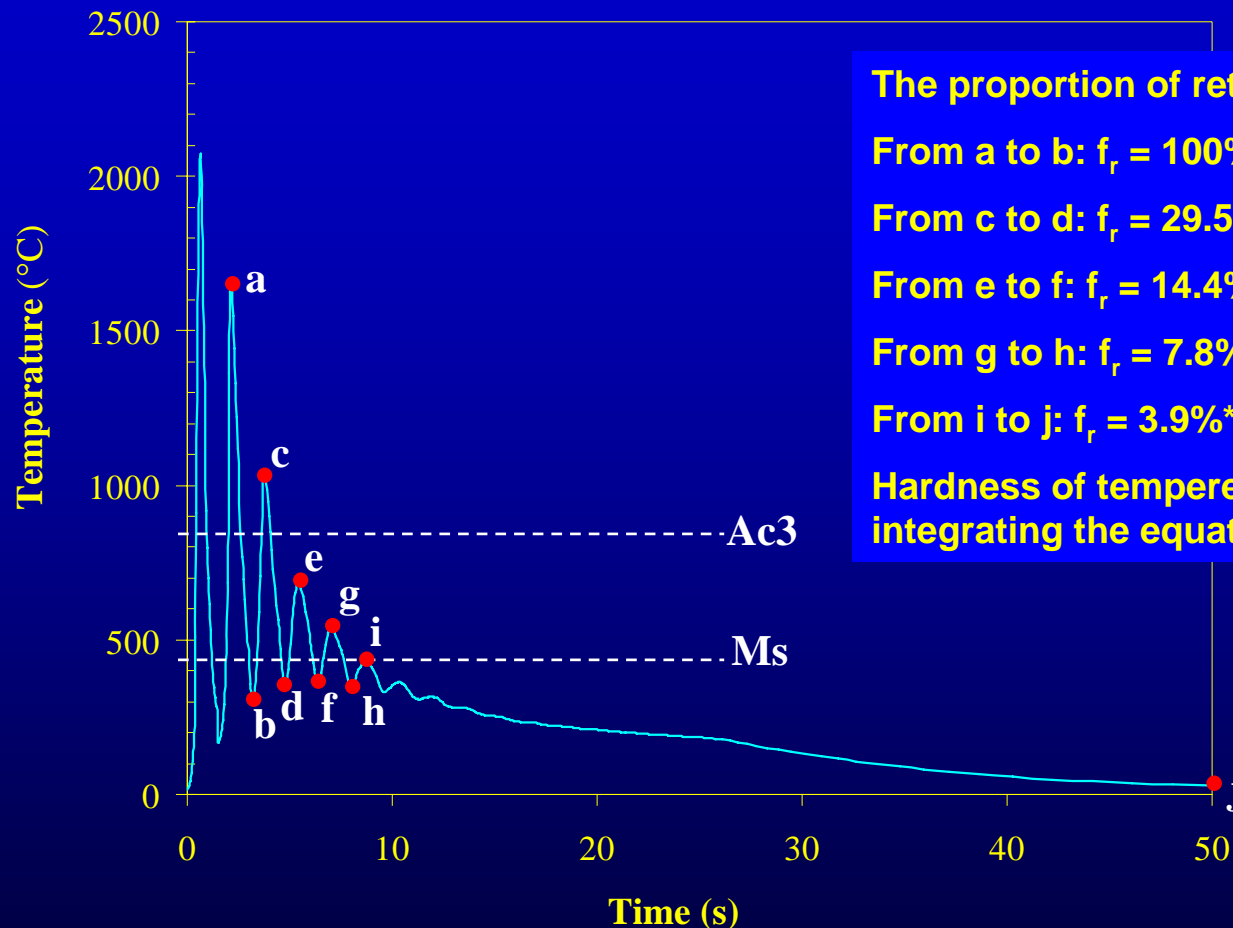


# Thermal Cycles



**Temperature profile at the center point of each layer**

# Thermal Cycle (1<sup>st</sup> layer)



The proportion of retained austenite is:

From a to b:  $f_r = 100\% \cdot \exp[-(420-309)/90.9] = 29.5\%$

From c to d:  $f_r = 29.5\% \cdot \exp[-(420-355)/90.9] = 14.4\%$

From e to f:  $f_r = 14.4\% \cdot \exp[-(420-364)/90.9] = 7.8\%$

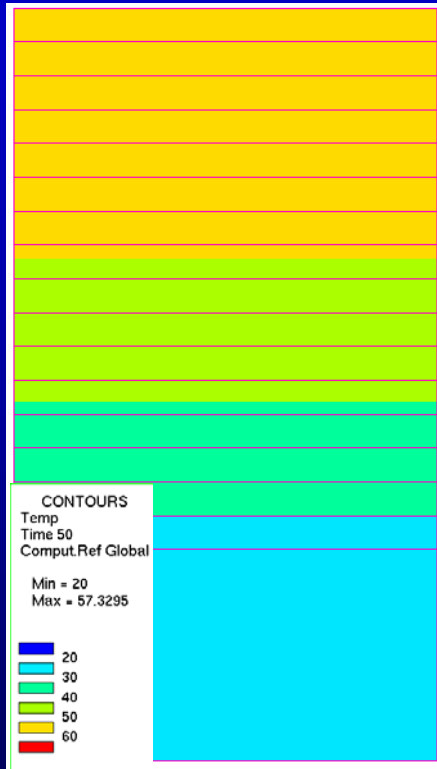
From g to h:  $f_r = 7.8\% \cdot \exp[-(420-358)/90.9] = 3.9\%$

From i to j:  $f_r = 3.9\% \cdot \exp[-(420-30)/90.9] = 0.0\%$

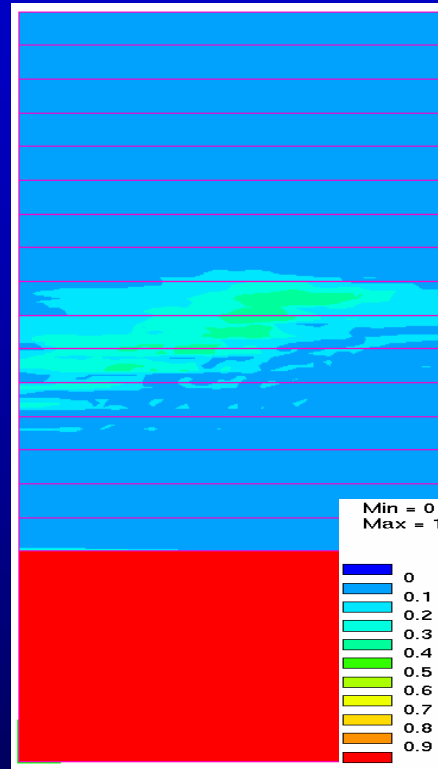
Hardness of tempered martensite is calculated by integrating the equation from point b to j.

- Martensite is completely tempered at the first layer ( $H_M = 582\text{HV}$ )

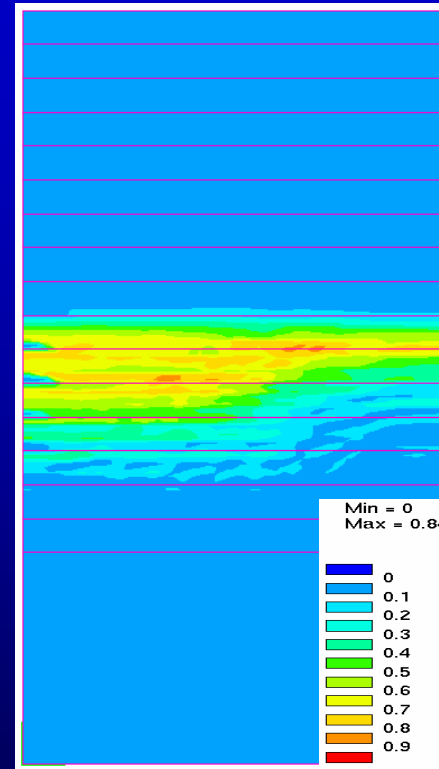
# Phase Contours (t=50s)



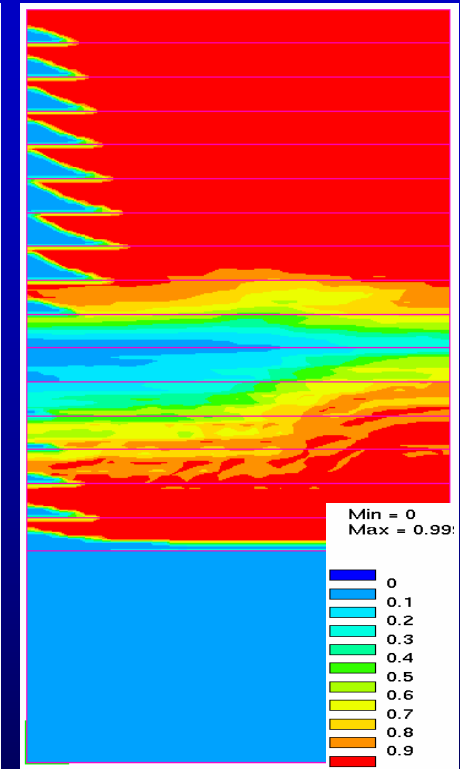
Temperature



Ferrite



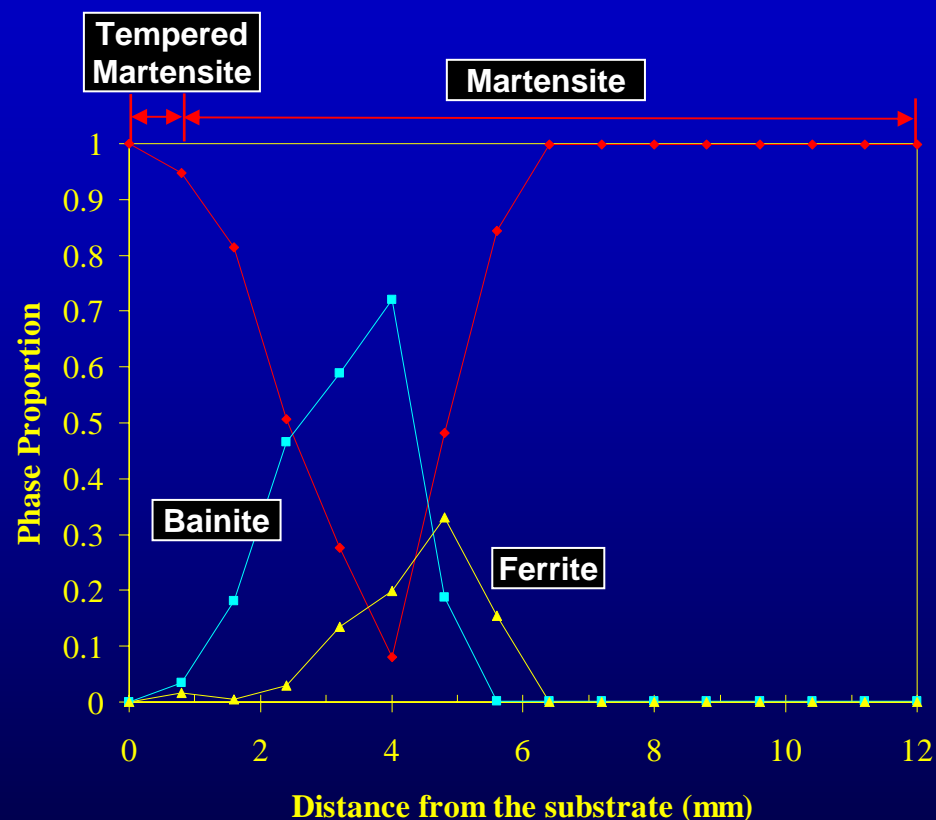
Bainite



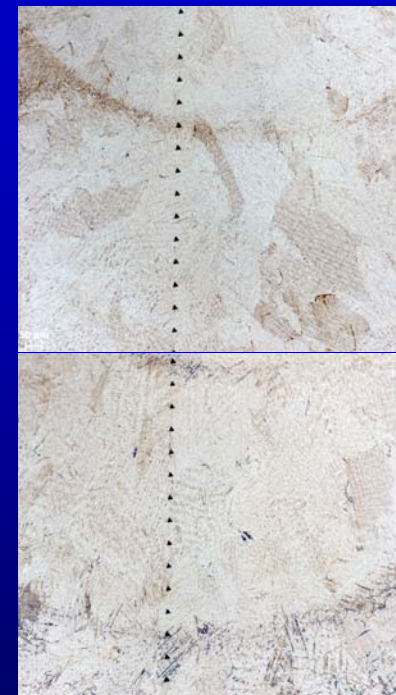
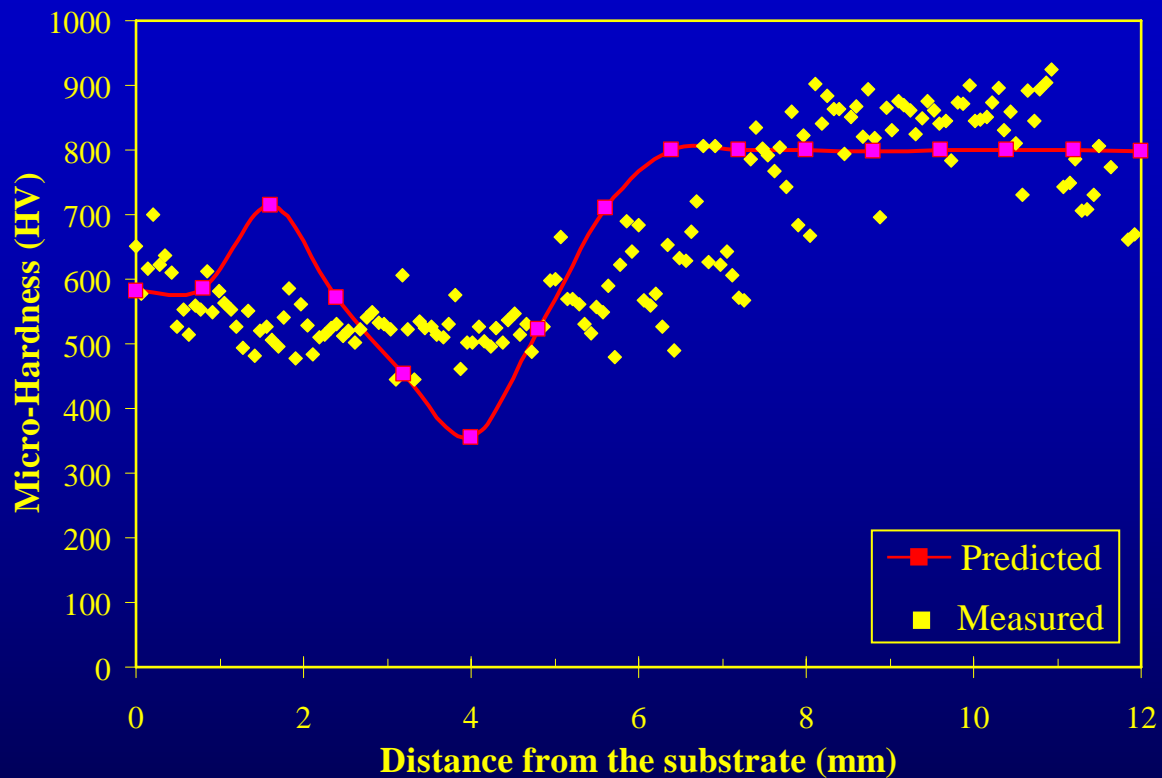
Martensite

# Phase Proportions

- The first 2 layers are tempered martensite since the part cools down below the martensite start temperature ( $M_s$ ) for each thermal cycle
- Starting from the third layer, the part never cools down to  $M_s$  and martensite is not tempered once it forms
- From the 3rd to 7th layers, bainite forms due to the slow cooling rate, then martensite forms afterwards with increasing cooling rate



# Micro-hardness Measurement and Prediction



# Conclusion

- 3-D model was developed to predict the thermal cycles, phase transformation, and hardness in LENS deposition process.
- The calculated hardness qualitatively agrees with the measured data
- The microstructure and hardness strongly depends on the thermal history in the deposited part.
- This model has the potential to control the process-property relationships for the component design optimization to meet the product attributes.